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Cu-ZnO Nanostructures Synthesis and Characterization

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Abstract

5wt% copper doped zinc oxide (Cu-ZnO) nanostructures were prepared via the hydrothermal technique at different temperatures of 70, 100, 130, 160 and 190°C. UV spectroscopy, FE-SEM microscopy, XRD crystallography, and EDS measurements were used for nanostructure characterization. UV spectroscopy indicated a red shift for the absorption peaks, and hence a blue shift for the energy gap values, as temperature increased from 70 to 190°C. FE-SEM microscopy showed an increase in the average lengths and diameters of the nanostructures following a similar increase in temperature. XRD crystallography indicated decent structural patterns for Cu-ZnO nanostructures with an increase in crystallite size upon temperature increase. Interestingly, three unprecedented extra indices appeared in the structural pattern at 190°C, which might indicate a configuration of hexagonal crystallite with three extra planes. EDS measurements indicated the sole presence of Cu, Zn and O.

Keywords: Nanostructures, Nanorods, Copper doped, Cu-ZnO nanostructures, Semiconductors, Hydrothermal.

تصنيع ووصف اوكسيد الزنك المشوب بالنحاس ذو التركيب النانوي

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الخلاصة

تم تصنيع المركبات النانوية لأوكسيد الزنك المشوب بالنحاس بنسبة وزنية 5% بواسطة طريقة تقنية الهيدروثيرم و بدرجات حرارية مختلفة (70، 100، 130، 160 و 190 مؤي). لتحديد مواصفات المتراكبات النانوية تم قياس النماذج باستخدام طيف الأشعة فوق البنفسجية، المجهر الإلكتروني الماسح، تشتت الأشعة السينية، التشتت الطافي لطيف الأشعة السينية. أظهرت نتائج قياسات طيف الأشعة فوق البنفسجية (UV spectroscopy) زيادة في الطول الموجي وانخفاض في قيم فجوة الطاقة مع ارتفاع درجة حرارة تحضير العينات (70-190 مؤي)، بينما بينت صور وقياسات المجهر الإلكتروني الماسح (FE-SEM) زيادة في أطوال واقطار المركبات النانوية بزيادة درجة حرارة التحضير. أظهرت قياسات الأشعة السينية انحراف قليل عن اوكسيد الزنك النقي في مواقع وارتفاع القمم عند 190°C بينما ظهرت ثلاثة قمم جديدة في 190°C وهي لم تكن مسجلة او مذكورة مسبقاً من قبل الباحثين باستخدام هذه التقنية. ونحن نعتقد بتكون بلورات سداسية مع ثلاث اوجه جديدة، وقد اشارت قياسات التشتت الطافي لطيف الأشعة السينية (EDS) لوجود O, Zn, Cu فقط.

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Introduction

Zinc oxide is an inorganic white powder with the chemical formula of ZnO and crystallite configuration of either wurtzite hexagonal or zinblende cubic. Wurtzite hexagonal is the most common configuration since it is more stable, whilst zinblende is less common since it requires certain conditions to be configured. In either way, zinc and oxygen are spatially arranged as tetrahedral centres [1-5]. As a II-VI semiconductor with wide band gap energy of 3.37 eV and large exciton binding energy of 60 meV, ZnO is being used for decades for various optical, electrical, and mechanical applications. These include piezoelectric devices, light emitting diodes (LEDs), gas sensors, photocatalysts in solar cells, corrosion inhibitors, and antibacterial biosensors [6-10]. Apart from *zincit*, which is a zinc oxide that is naturally found with hexagonal configuration, several methods are being used to produce ZnO as either bulk particles, e.g. via French process, indirect method, and wet chemical process, or as nanostructures, e.g. via hydrothermal technique, sol-gel method, laser ablation, etc. [2-4, 6-12]. In the present work, we used the hydrothermal technique since it requires low temperature, low expenses, and short time, being environmentally clean and nontoxic, with key parameters that could be easily tuned and controlled [2, 3, 12]. As we already suggested in our previous work, defects may have great potentials to enhance optical, electrical, mechanical, and chemical properties [3]. Nevertheless, defects do not only genuinely occur, but could also be deliberately induced via doping in order to modify the crystallite structure. We and others believe that the latter approach would have useful potentials [13-18]. Doping might modify the crystallite tetrahedral configuration of ZnO and, hence, affect its entire corresponding properties. For example, if ZnO is doped with a transitional element, such as copper, nickel, or cobalt, its luminescence capacities would be enhanced and, hence, it could be used in bioimaging for sensitive detection. Another example can be the doping of ZnO with copper, where conductivity would be increased since copper is a highly conductive metal and, hence, it could be altered from a semiconductor to a conductor [11, 12, 19-24]. In this study, we doped 5wt% copper within ZnO crystallite structure using the hydrothermal technique at different temperatures of 70, 100, 130, 160 and 190°C. The doped structure was then characterized using UV spectroscopy, FE-SEM microscopy, XRD crystallography, and EDS measurements. We aimed to show the effects of temperature variation on the shapes and sizes of Cu-ZnO nanostructures.

Experimental Part

Methodology

Cu-doped ZnO (Cu-ZnO) nanostructures were prepared at a range of temperature of 70, 100, 130, 160 and 190°C using 1M zinc acetate dihydrate ($\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$, M.W=219.5 $\text{g}\cdot\text{mol}^{-1}$, BDH Chemicals), 5M sodium hydroxide (NaOH, MW=40.99, 98%, ALPHA) and 5% copper acetate dihydrate ($\text{Cu}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$, M.W=217.5 $\text{g}\cdot\text{mol}^{-1}$, Science Company). All solutions were prepared using 95% methanol (CH_3OH , MW= 32.04 $\text{g}\cdot\text{mol}^{-1}$, Scharlau) as a solvent. The hydrothermal technique was used in order to maintain high temperature and pressure via a Teflon-lined stainless steel autoclave system (polypropylene liner PPL). The reactor with a curled heater was encircled in order to maintain equal and continuous heat distribution during the reaction (Figure-1). 5wt% of $\text{Cu}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$ was dissolved in 10ml of 1M $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$ and well mixed via magnetic stirring for five minutes before mixing with 40ml of 5M NaOH inside the reactor. The reactor was then sealed and left for at least 10 hrs at the above-mentioned temperatures. The mixture was then decanted in a 100 ml beaker and washed several times with distilled water until reaching a pH value of 7. Then, a final washing step with ethanol was performed for at least three times. The nanostructures powder was then separated and dried under 100°C for 2 hrs.

Instrumental Tests

The synthesized samples were characterized for their structure by x-ray diffraction (SHIMADZU 6000/X-Ray / Japan.) with Cu $K\alpha$ radiation. The optical band gap (E_g) was estimated via the UV-Vis-NIR diffuse reflectance spectroscopic (SHIMADZU 1800/ Double Beam/ Japan) studies in a wavelength range of 200 nm to 1100 nm. The samples were used in the form of powder, while FE-SEM (S-4160/Hitachi/ Japan) was used as the reference for images. The elemental composition of the ZnO nanostructures was determined by using EDS S-4160/Hitachi/Japan.



Figure 1- Hydrothermal system consisting of two main parts: (a) Temperature controller unit.(b) Reaction unit.

Results and Discussion

XRD crystallography and EDS measurements

XRD crystallography indicated the purity and well crystallinity of Cu-ZnO and the undoped ZnO nanostructures with hexagonal wurtzite crystallites, as they all matched the structural pattern for pure zinc oxide described by JCPDS (Joint Committee on Powder Diffraction Standards) card no 01-079-0205. At two different temperatures of 70 and 190 °C, XRD crystallography showed intense diffraction angles for the indices of 100, 002, 101, 102, 110, 103, 200, 112, 201, 004, and 202 at a range of diffraction angles of 10 to 80 degree (Figure-2). Crystallite size was estimated by applying Scherrer equation, as previously described [3]. As demonstrated in Figure-2, crystallite size of ZnO nanostructures was increased from 19.44 to 30.55nm whereas that of undoped ZnO nanostructures was increased from 21.75nm to 37.05 nm as temperature was increased from 70°C to 190 °C, respectively. Also, peak intensities for Cu-ZnO nanostructures were found to be reduced in comparison to those for undoped ZnO nanostructures, which obviously indicated a reduction in crystallite size [3]. Based on these observations, the crystallite size of Cu-ZnO nanostructures is considered to be decreased in comparison to that of undoped ZnO nanostructures, which is expected and in line with results reported by other studies, since doping with copper implies that the zinc atom would be replaced by the much smaller atom of copper [24-27]. Interestingly, at 190°C, Cu-ZnO diffraction pattern showed three new peaks at 43.46, 50.62, and 74.30°, which, as far of our knowledge, have not been previously reported. This might indicate the formation of hexagonal crystallites with three extra planes (Figure-2). In fact, one might argue that these extra peaks could reflect crystallite impurity, but EDS measurements indicated the sole existence of Cu, O, and Zn in Cu-ZnO nanostructures and O and Zn in the undoped ZnO nanostructures (Figure-3). At this stage, it is not possible to define the exact configurations of these extra planes neither the exact parameters which might affect their growth. It is even not possible to define the appearance of yet more planes, or how these extra planes have modified the spatial configuration of the crystallites, and how their optical, mechanical, electrical, and chemical properties would be changed accordingly. It is a matter of interest to resolve these issues in future works.

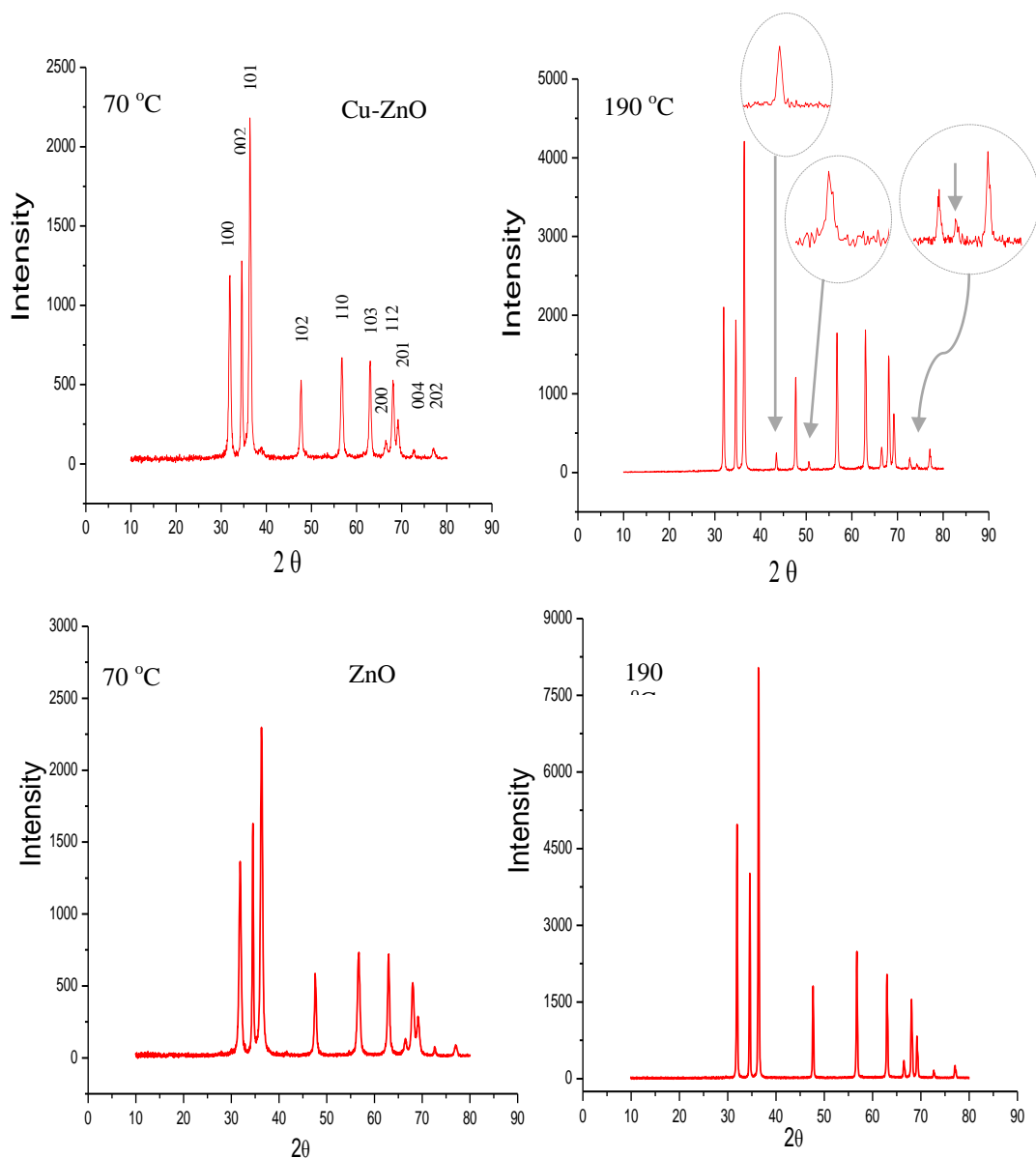


Figure 2- XRD crystallography for Cu-ZnO nanostructures (top) and undoped ZnO nanostructures (bottom) at temperatures of 70 and 190 °C. The patterns indicate a wurtzite hexagonal configuration which perfectly match the pattern given by the Reference code:01-079-0205. At 190°C (top), we emphasized the three extra peaks at 43.46, 50.62 and 74.30° via three inset spectra.

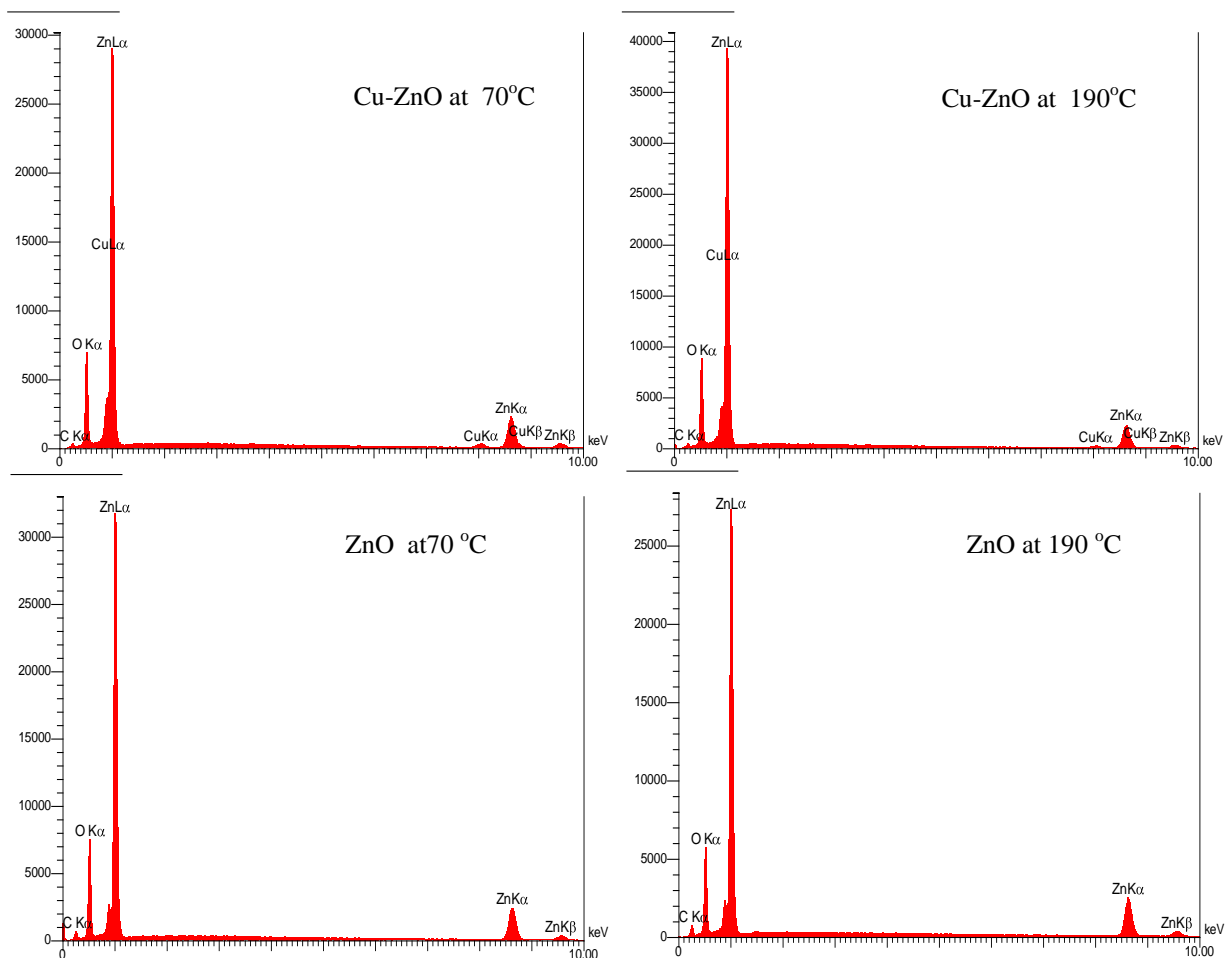


Figure 3-EDS spectra for Cu-ZnO nanostructures (top) and undoped ZnO (bottom) synthesized via the hydrothermal technique at 70 and 190 °C.

FE-SEM microscopy

FE-SEM microscopy showed an obvious temperature impact on the growth and morphology of the Cu-ZnO nanostructures. This is consistent with our previous work and that of others [28-31]. We believe that the growth and, hence, the change of particle size and shape, should be expected as temperature or incubation time, or maybe both, are increased. This is because of a natural phenomenon called *Ostwald Ripening*, in which small particles with small surfaces, and, hence, high kinetic energy, start to deposit themselves on the surfaces of the larger-sized particles then diffuse within these particles to end up with yet larger ones. In other words, this phenomenon occurs to stabilize the small particles [2, 3, 32-34]. Figures-4 and 5 show FE-SEM images for Cu-ZnO nanostructures and the undoped ZnO nanostructures synthesized at two different temperatures of 70 and 190°C, respectively, and at three different scales bars of 200 nm, 500nm and 1 μ m. In Figure-4(a,b,c), the average size of Cu-ZnO nanostructures was found to be 20-50 nm, with polygonal dots-like shapes. In Figure-4(d,e,f), the average size of the undoped ZnO nanostructures was found to be 150-200 nm, with rod like shapes. The average length was 150-200 nm and the average diameter was 10-20 nm. In Figure 5(a,b,c), the average size of Cu-ZnO nanostructures was found to be 200-300 nm with a variation from trigonal to hexagonal rod-like shapes. In Figure-5(d,e,f), the average size of the undoped ZnO nanostructures was found to be $>2\mu$ m (the average length was 200-2000 nm and the average diameter was 100-1000 nm), with tetrahedral-like shapes. In general, Figures- 4 and 5 show that the average size of Cu-ZnO nanostructures is much smaller than that of the undoped ZnO nanostructures, which is

in line with results from earlier works [35-37]. These results are also in line with the results of our XRD crystallography, which showed smaller crystallite size of Cu-ZnO nanostructures in comparison to that of the undoped ZnO nanostructures (Table-1).

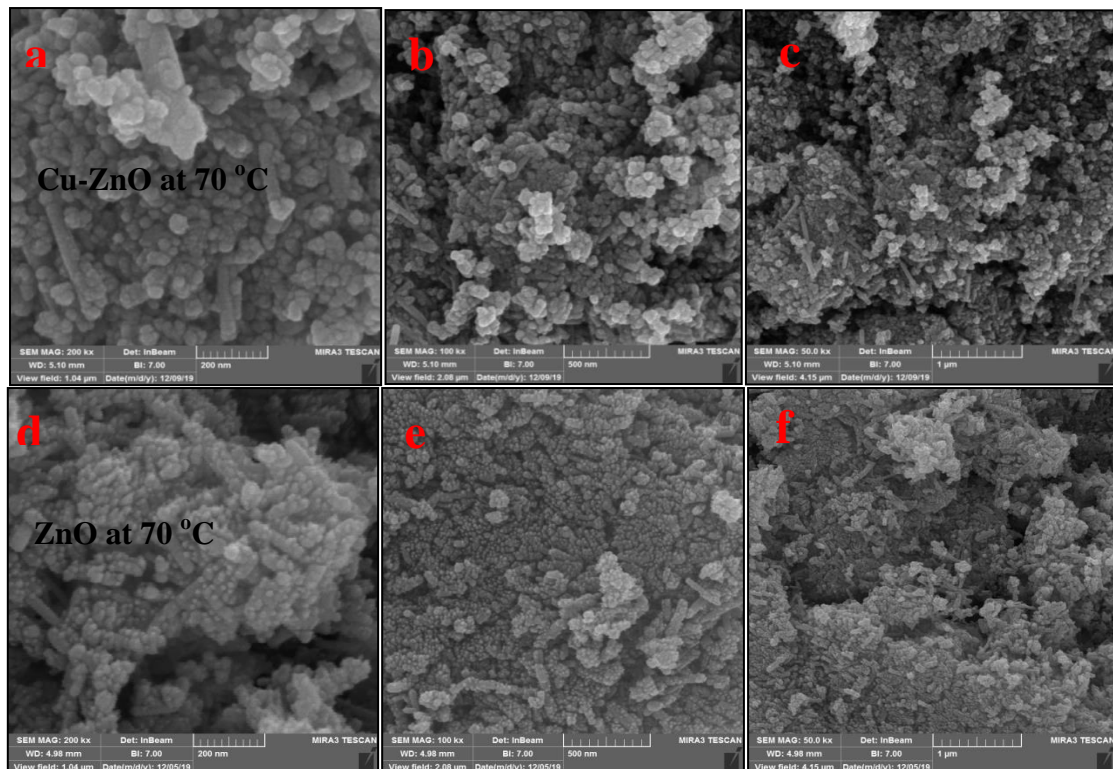


Figure 4- FE-SEM images for Cu-ZnO and undoped ZnO nanostructures synthesized at 70 °C using Teflon lined stainless steel autoclave in a hydrothermal system. Images are shown in three scales, namely: a. 200nm, b. 500nm and c.1 μ m for Cu-ZnO nanostructures and d. 200nm, e. 500nm and f.1 μ m for undoped ZnO nanostructures.

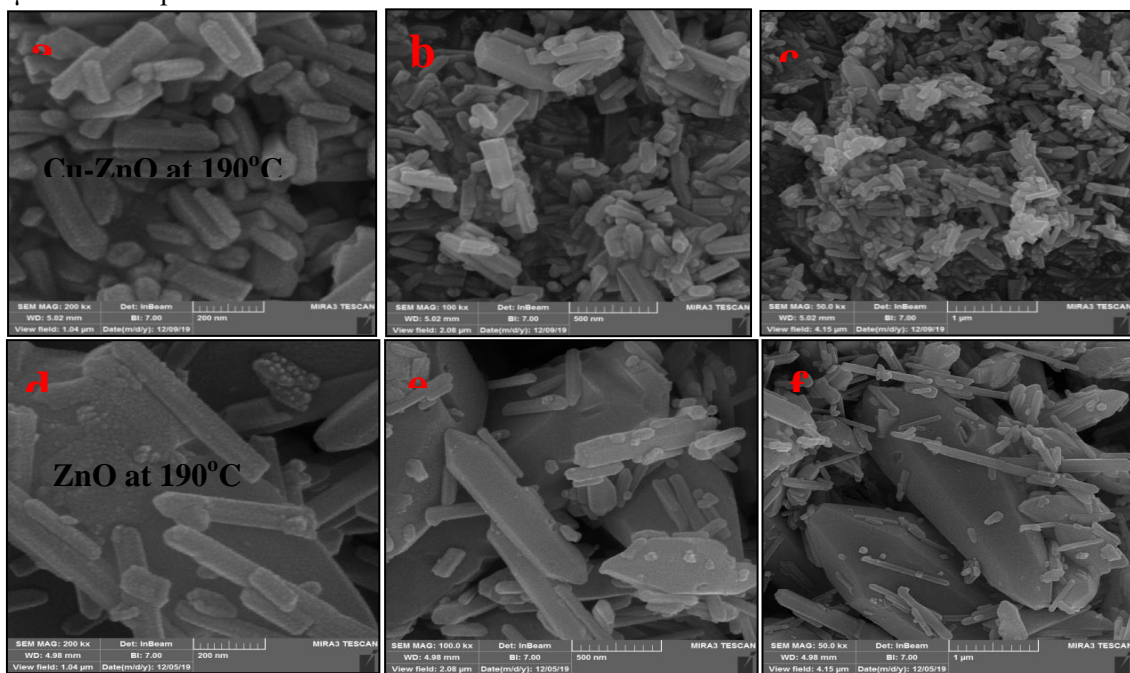


Figure 5- FE-SEM images for Cu-ZnO and undoped ZnO nanostructures synthesized at 190 °C using Teflon lined stainless steel autoclave in a hydrothermal system. Images are shown in three scales, namely: a. 200nm, b. 500nm and c.1 μ m for Cu-ZnO nanostructures and d. 200nm, e. 500nm and f.1 μ m for undoped ZnO nanostructures.

Table 1- Particle and crystallite sizes for Cu-ZnO and undoped ZnO nanostructures determined by SEM microscopy and XRD crystallography.

Nanostructures	Cu-ZnO		ZnO	
	70 °C	190 °C	70 °C	190 °C
Temperature				
Average particle shape	Polygonal dots	Trigonal and hexagonal rods	Rod like shapes	Tetrahedral
Average particle size via FE-SEM microscopy	(20-50) nm	(200-300) nm	(150-200) nm	(200-2000) nm
Average crystallite size via XRD crystallography	19.44 nm	30.55 nm	21.75 nm	37.05 nm

UV-VIS spectroscopy

UV-VIS spectra showed absorption peaks at 370, 372, 373, 375, and 377 nm for Cu-ZnO nanostructures synthesized at five different temperatures of 70,100,130,160, and 190°C, respectively (Figure-6a). Following the approach utilized by a previous study [2], the values of band gap energy E_g were determined. The following equation was applied:

$$E_g = hc/\lambda$$

where h is the Plank constant= 4.135667×10^{-15} eV.s, C is the speed of light= 3.00×10^8 m/s, and λ is the maximum wavelength of the absorption peaks in nm.

The values of E_g were found to be 3.35324, 3.33522, 3.32627, 3.30853, and 3.29098eV at the temperatures range of 70,100,130, 160, and 190°C, respectively. For comparison purposes, the results in Figure-6b for the undoped ZnO nanostructures synthesized using exactly the same parameters showed absorption peaks at 355, 360, 370, 371, and 373 nm, respectively, at the same temperature values. The values of E_g were found to be 3.49493, 3.44639, 3.35324, 3.34421 and 3.32627eV, respectively, at the same temperatures range. In general, E_g values were found to be decreased as the wavelength of absorption peaks and temperatures were increased (Figure- 6c,d and Table-2). This is because particle size naturally tends to be increased as temperature is increased [2, 3]. In Table-2, we show a comparison between the band gap energy values for Cu-ZnO nanostructures and those for the undoped ZnO nanostructures to show the effects of copper doping on the absorption peaks of the wavelengths and, accordingly, on the corresponding band gap energy values. Generally speaking, Cu-ZnO nanostructures showed higher absorption peaks and, hence, lower band gap energy values in comparison to undoped ZnO nanostructures, which is in line with the results of other previous works [35, 36, 38]. However, this effect is not due to the particle size as one would expect when examining the UV spectra. As it already mentioned above, XRD crystallography and FE-SEM microscopy showed an obvious reduction in the sizes of Cu-ZnO crystallites and nanostructures in comparison with those of undoped ZnO (Table-1). As such, it is our opinion that the reduction in band gap energy values is due to the free electrons in the outer shells of copper atoms, which are already conductive. We believe that this is a great property to be induced in ZnO nanostructures, since it would not only reduce the band gap energy values of ZnO as a semiconductor, but it would also reduce the size of ZnO nanostructure, i.e. increase its surface area. For this reason, we suggest that copper doping might highly enhance the conductivity, since copper is a conductive metal, and also enhance the optical activity, since it reduces the band gap energy values of ZnO nanostructures. As such, we expect that Cu-ZnO nanostructures would be an inexpensive and powerful candidate that one would choose to

enhance the solar cells efficiency, or to enhance

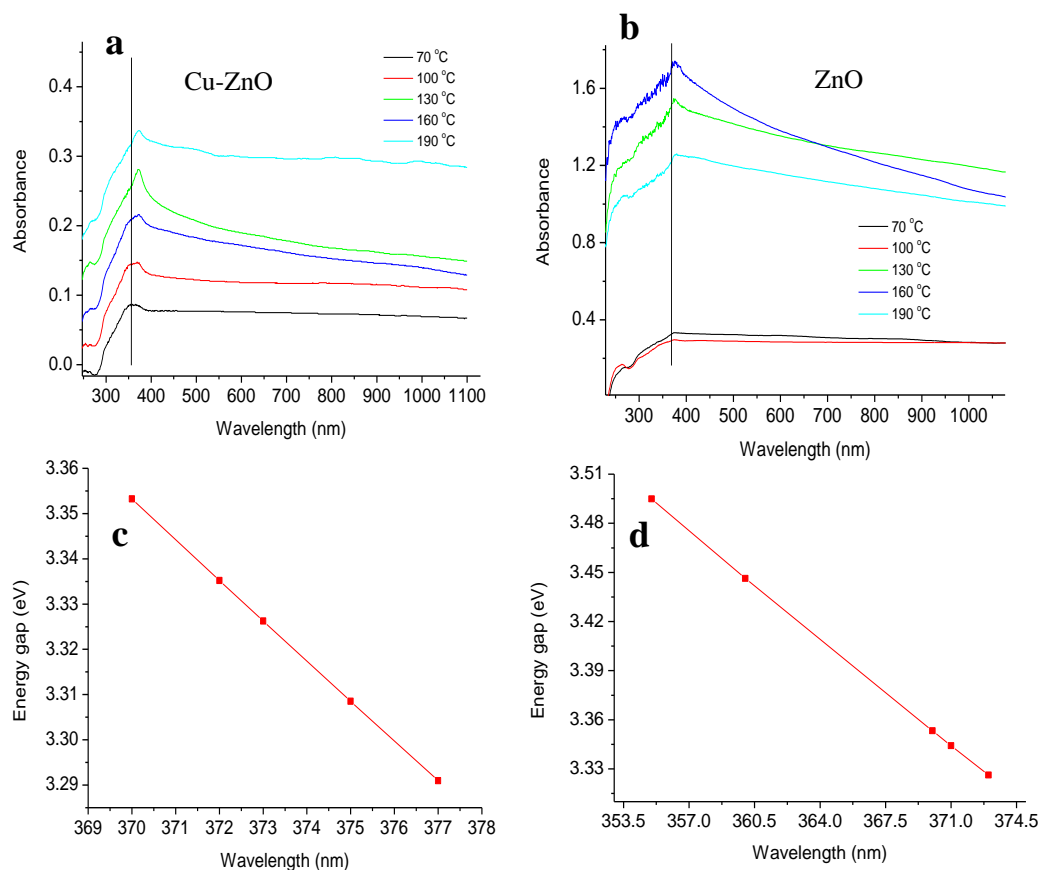


Figure 6-Top row: UV-Vis spectra at the region of 225-1100 nm for (a) Cu-ZnO nanostructures and (b) undoped ZnO nanostructures; both prepared at different temperatures of 70, 100, 130, 160 and 190 °C. Bottom row: the relationship between band gap energy values and the corresponding wavelength absorption peaks for (c) Cu- ZnO nanostructures and (d) undoped ZnO nanostructures, taken from a and b.

since the green luminescence which could be used for high bio-imaging performance, i.e. for highly sensitive detection of pathogens.

Table 2- Comparison of absorption peaks and band gap energy values for Cu-ZnO and undoped ZnO nanostructures.

Synthesis Temperature (°C)	Cu-ZnO		undoped ZnO	
	Wavelength (nm)	Energy gap (eV)	Wavelength (nm)	Energy gap (eV)
70	370	3.35324	355	3.49493
100	372	3.33522	360	3.44639
130	373	3.32627	370	3.35324
160	375	3.30853	371	3.34421
190	377	3.29098	373	3.32627

Conclusions

The hydrothermal technique was found to be a low-cost, and short-time based technique with low temperature requirements, which could be used to produce Cu-ZnO nanostructures with appreciable amounts of pure and well configured crystallites. 5wt% copper was found to be sufficient to reduce ZnO crystallite size and, interestingly, to modify ZnO crystallites into a new hexagonal configuration

with three unprecedented extra planes which we believe that they are either outer planes, inner planes, or possibly both. The exact scenario of this configuration would be an attractive topic for future work. Maintaining higher temperature is a preferable routine to be followed in order to increase nanostructure size, due to Ostwald Ripening phenomenon.

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